

## Microstructural change in ice: II. Creep behavior under triaxial stress conditions

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**ABSTRACT.** This work investigates the deformation of ice under deviatoric stresses and confining pressures expected during ice–structure interaction. Granular ice was tested under a range of confining pressures (5–60 MPa) and deviatoric stresses (up to 25 MPa), with sample temperatures between  $-8^{\circ}$  and  $-10^{\circ}$ C. Samples were deformed to increasing end-levels of axial strain, and were thin-sectioned and photographed immediately following testing.

At all confinement levels, the original texture of the sample is completely transformed within the first 10–15% strain, to a fine-grained matrix with a few larger, isolated grains. At low confinements, grain-size reduction is associated with extensive microcracking. At high confinements, few cracks are observed. Observations suggest that microcracking, melting and recrystallization are active at all levels of confinement, though the relative importance of each depends on the confinement, stress and accumulated strain.

Deviatoric stress is a strong factor in controlling the creep, reflected in both the time required for the sample to reach accelerated creep and the tertiary creep rate itself. Two exceptions to this pattern were noted. First, some samples experienced strain localization and eventual rupture. These specimens tended to have higher creep rates even in the initial stages of strain. Second, prior damage resulted in rapid softening compared with the behavior of undamaged specimens. However, when strain rates are compared among all samples at a given level of cumulative axial strain, the creep behavior obeys a power law over the whole range of strain levels tested. Effective viscosity decreased from  $10^{7.8}$  to  $10^{6.4}$  MPa<sup>-n</sup>s within the first 10% strain, during which the most substantial microstructural changes occurred, and then stayed relatively stable. The stress exponent,  $n$ , remained within the range 4.0–4.6.

The dominant deformation mechanism appears to depend strongly on confining pressure (cracking at low pressure and dynamic recrystallization at high pressure). Creep rates at high confinement appear to increase relative to those at intermediate confinement, but the influence of temperature must be addressed further.

### INTRODUCTION

Much attention to the mechanical behavior of ice has focused on the slow strain rates, low confining pressures and small deviatoric stresses associated with deformation within glaciers. Ice mechanics applied to ice–structure interaction introduces new challenges both for experimental studies and for constitutive modeling of ice behavior. The extreme interaction rates, high shear stresses, high confining pressures, rapid and continuous damaging of the ice during impact and the occurrence of multiple deformation mechanisms within a contact zone make the problem extremely complex. We have focused on two objectives: (1) laboratory studies of the critical factors involved in ice failure under the range of conditions acting in an impact zone, and (2) the development of a constitutive relation that will describe the complex behavior of ice under these conditions.

The studies presented in this and the two companion papers (Melanson and others, 1999; Muggeridge and Jordaan, 1999) are primarily aimed at addressing the first objective and

using the results to guide our approach to the second. The experimental work presented in Melanson and others (1999) and this paper was not undertaken to reproduce impact conditions. Rather, the approach was to isolate two of the critical elements — high-stress conditions and the damage-formation process — and systematically characterize the mechanical and microstructural response of ice to these factors. In Muggeridge and Jordaan (1999), iceberg ice subjected to true impact conditions in field tests is characterized.

Measurements of loads during ship-ramming trials, indentation tests and laboratory compression tests indicate ice can sustain, and therefore transmit, very high loads during impact. Pressures on the order of 70 MPa have been documented in indentation tests for a sensor of diameter 12.7 mm (Frederking and others, 1990); pressures up to 51 MPa were detected on an instrumented ship hull for a sensor of area  $0.0472\text{ m}^2$  (Glen and Blount, 1984). These high loads occur within localized areas termed “high-pressure zones”. The ice is able to sustain these loads in part due to the confining action of the surrounding ice mass. Extensive damage to the ice occurs rapidly (typically within a few seconds or less), and extrusion of pulverized ice is commonly observed. Large deformations are accommodated by intense damage within a

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localized zone adjacent to the impact surface. Observations of ice subjected to impact tests indicate that fracturing, recrystallization and melting are deformation mechanisms acting within an impact zone (Kheisin and Cherepanov, 1973).

The term “damage” has typically been used to define microstructural elements (such as microcracks) which accumulate in a material under sustained stress and cause the degradation of properties such as modulus or strength. In the context of the current research there are a number of potential damage elements, including microcracks, dislocations and melt fraction, that can act to soften or weaken the ice. A key objective of this work is therefore to understand the microstructural changes associated with the change in mechanical behavior resulting from damage. For further discussion of damage to ice during ice–structure interaction, see Jordaan and others (1999) and Melanson and others (1999).

While extensive experimental work has been done on many types of ice under uniaxial and biaxial conditions (Sinha, 1979; Cole, 1983; Mellor and Cole, 1983; Jacka and Maccagnan, 1984; Smith and Schulson, 1993), fewer studies have addressed the deformation and failure of granular ice under triaxial confinement (Jones and Johari, 1977; Jones, 1982; Jones and Chew, 1983; Rist and others, 1988; Stone and others, 1989; Kalifa and others, 1992; Rist and Murrell, 1994; Rist and others, 1994; Gagnon and Gammon, 1995; Stone and others, 1997). Because high confinements, high shear stresses and large strains occur in impact zones during ice–structure interaction, the present study was undertaken to assess the mechanical behavior and associated microstructural changes in ice under these stress conditions. Laboratory test conditions covered a range of confinements from low pressures (present at the periphery of the high-pressure zone) to high pressures (found at the center of a high-pressure zone where the ice is either relatively intact or is highly damaged and constrained by the surrounding ice). A range of deviatoric stresses from 2 to 25 MPa was used; microstructural characterization focused on samples tested using a deviatoric stress of 15 MPa.

## EXPERIMENTAL PROCEDURES

### Sample preparation

Cylindrical test specimens of granular ice measuring 70 mm in diameter and 155 mm in length were made using the procedure described in Melanson and others (1999). In order to compare the effects of initial grain-size on ice behavior, one sample was made from a block for which snow, rather than crushed ice, was used as seed material to produce a finer-grained ice. Another sample was prepared from one of the large-grained blocks typically used for crushing into seed.

### Test procedure

Constant-axial-load tests were done using a Materials Testing Systems (MTS) test frame with independent servo-controlled actuators for applying axial load and confining pressure. Load paths were programmed using MTS Test-Star software. Samples were jacketed in latex to exclude the confining medium (silicon oil) and placed in the triaxial confining cell. The cell was filled and a small axial load was applied to ensure contact between the loading piston and the specimen. Then the confining pressure was increased at a rate of  $0.5 \text{ MPa s}^{-1}$  and held for approximately 1 minute prior to the creep-load step. The creep load was

applied rapidly (the target load was typically reached within 0.1 s) and then held constant as the sample deformed. Axial load was monitored on two load cells located inside and outside the triaxial vessel. Strain was not monitored directly on the specimens, because the apparatus could not accommodate the large sample distortions. Displacement of the loading piston was therefore used to monitor the amount of sample shortening. The axial load was quickly removed as soon as the specified strain limit was reached, and then the confining pressure was slowly decreased.

In general, samples were allowed to deform until a prescribed limit of axial strain was reached. Most were taken to the maximum limit (approximately 44% true strain). For two values of confining pressure (5 and 50 MPa), a number of tests were halted at several intermediate levels of strain in order to examine the progressive microstructural changes occurring.

The point of contact with the piston could generally be determined to within 0.4 mm at the start of the creep step; for a sample taken to 4% strain, this introduces uncertainties on the order of 5–6% of the total strain measurement (or approximately 0.2% strain). For the tests taken to the maximum strain limit, it reflects an uncertainty of  $< 1\%$  of the total strain measurement. For this reason we have not focused on the behavior of the samples at very small strains in the tests presented here; they were not designed for that purpose.

The test samples, confining oil and test apparatus were held at a temperature of  $-10^\circ\text{C}$  for at least 12 hours prior to testing. Work by B. M. Stone (unpublished data, 1995) and subsequent calibration experiments by Melanson (1998) indicate that the application of confining pressure causes an increase in temperature at the center of an ice specimen of approximately  $0.02^\circ\text{C MPa}^{-1}$  and an increase in the temperature of the confining oil of approximately  $0.11^\circ\text{C MPa}^{-1}$ . In the tests done at the highest confining pressures, attempts were made to compensate for this shift in temperature by using a slightly colder initial sample/oil temperature of  $-11^\circ$  to  $-12^\circ\text{C}$ . More precise control over the sample temperature was not possible given the configuration of the test apparatus.

### States of stress and damage

The primary focus of testing was to study the creep behavior of initially intact ice under a range of triaxial confinements with high deviatoric stresses. A number of additional stress paths were also investigated to assess the effects of deviatoric stress, prior damage and the effects of a rapid drop in confinement which would accompany ice failure in a contact zone. In our notation, the applied confining pressure is designated  $P_c$ . The deviatoric stress (equivalent to the von Mises stress in this case, i.e. the difference between the axial stress and the applied confining pressure) is designated  $s$ .

#### *Creep tests on intact ice*

The majority of creep tests were done on initially intact ice using a nominal deviatoric stress ( $s$ ) of 15 MPa with confining pressures ( $P_c$ ) of 5–60 MPa (Table I). Tests were done over the range of confinements to assess the pressure dependence of creep at a constant deviatoric stress. To investigate the deviatoric stress dependence of creep, a series of tests was done in which  $s$  ranged from 2 to 20 MPa and  $P_c$  was 20 MPa. A series of tests aimed at characterizing the sample behavior for a constant hydrostatic stress ( $p$ ) and different deviatoric stresses is also presented. The hydrostatic stress,  $p$ , is defined under these test conditions as  $p = s/3 + P_c$ . (This parameter is

Table 1. The maximum axial strain (in % true strain) for each test specimen, with the confining pressure ( $P_c$ ) and deviatoric stress ( $s$ ) during the creep phase of the test

$P_c$ MPa	$s$ MPa	True strain											
		%	%	%	%	%	%	%	%	%	%	%	
5	15	4.2	10.7	18.0 <sup>1</sup>	21.6	29.6 <sup>2</sup>	44.4						
15	15	8.1											
20	2	1.7											
20	7	12.2											
20	20	17.4	34.4 <sup>4</sup>										
30	15	43.9	43.9 <sup>3</sup>										
30	25	2.5 <sup>4</sup>											
40	15	44.0 <sup>3</sup>											
45	15	39.5 <sup>4</sup>											
49	18	32.4 <sup>4</sup>											
50	15	2.4 <sup>5,4</sup>	4.1	10.5	16.4	29.9	32.5 <sup>4</sup>	35.0 <sup>4</sup>	39.7 <sup>1,4</sup>	41.7 <sup>4</sup>	44.1	44.1 <sup>6</sup>	
52	9	43.8											
53	6	43.8											
60	15	38.4 <sup>4</sup>											

<sup>1</sup> Sample tested at  $P_c = 5$  MPa to 18% strain, then  $P_c = 50$  MPa to 39.7% total axial strain.  
<sup>2</sup> Differential stress applied by dropping confinement.  
<sup>3</sup> Piston made hard contact with these samples during the initial application of the creep load, possibly damaged. Noted with tilde in the plots.  
<sup>4</sup> Samples ruptured; strain is approximately the point at which load dropped abruptly. Noted with asterisk in the plots.  
<sup>5</sup> Using large-grained sample. Noted "lg" in the plots.  
<sup>6</sup> Using fine-grained sample. Noted "fg" in the plots.

used primarily in the constitutive modeling (Melanson, 1998; Jordaan and others, 1999). In these tests, the applied confining pressure ( $P_c$ ) was adjusted such that the hydrostatic pressure was held at  $p = 55$  MPa for all the tests.

Finally, one specimen was subjected to an axial loading resulting in part from stress relief. In this test, a confining pressure of 30 MPa was applied and the piston was brought into contact with the sample. Then the confining pressure was dropped rapidly while the high axial load was maintained, resulting in an increase in deviatoric stress.

*Creep tests on damaged ice*

Because of the demonstrated difference in mechanical behavior between intact and damaged ice (Stone and others, 1997), the impact of prior damage on the creep behavior of ice was investigated. In these tests, samples were subjected to axial strains of either 2%, 4% or 12% at a constant rate of either  $10^{-2} \text{ s}^{-1}$  or  $10^{-4} \text{ s}^{-1}$  under 20 MPa confinement, prior to application of the creep load (Table 2). (For further details of these tests see Melanson, 1998.) In one additional test, the sample was subjected to 18% strain under creep conditions  $s = 15$  MPa and  $P_c = 5$  MPa. The confinement was then increased to 50 MPa and the sample was allowed to continue deforming at  $s = 15$  MPa until the maximum strain limit was reached (see Table 1, footnote 1).

*Stress correction for lateral strain*

For the tests taken to large strains, the increasing cross-sectional area resulting from lateral strain caused a decrease in the actual deviatoric stress applied under constant-load conditions. A correction has been applied to the measured loads to estimate the true stress, as discussed in Melanson and others (1999). These values are designated in our figures as "corrected deviatoric stresses".

**Sample analysis**

Samples were removed from the confining vessel, and then thin-sectioned and photographed within 24 hours of testing to minimize the influence of static recrystallization or crack

healing. Most of these samples were analyzed within 2 or 3 hours of testing. Although the possibility of crack healing is always a concern, samples which were significantly micro-cracked in thin section were, without exception, visibly cracked immediately upon removal from the confining cell. Samples taken to the higher strain levels were, in fact, opaque. Samples which were predominantly clear and transparent upon removal from the cell had correspondingly few cracks in thin section. Occasionally the presence was noted of a few larger (i.e. grain-sized) cracks immediately adjacent to the loading platens; this is a region in which the stress distribution is highly non-uniform, and thin sections were therefore not taken close to the platens.

Thin sections were cut perpendicular and parallel to the sample axis (designated as horizontal and vertical sections, respectively) and polished on both sides using a microtome, to a final thickness of 0.5 mm ( $\pm 0.1$  mm). To preserve the mi-

Table 2. Eight specimens were subjected to a damage step at the constant strain rate given, under confinement of 20 MPa. Following this step, a deviatoric stress ( $s$ ) was applied with confining pressure ( $P_c$ ), and the sample was allowed to creep to the maximum axial strain shown. These tests are noted with "d" in the plots

True strain	Damage step			Creep step	
	$P_c$	Strain rate	$P_c$	$s$	True strain
%	MPa	$\text{s}^{-1}$	MPa	MPa	%
2	20	$10^{-4}$	20	8	11.9
4	20	$10^{-4}$	20	20	34.2 <sup>1</sup>
12	20	$10^{-4}$	15	15	43.9
12	20	$10^{-4}$	30	15	43.9
12	20	$10^{-4}$	50	15	44.1
12	20	$10^{-2}$	15	15	44.0
12	20	$10^{-2}$	30	15	43.9
12	20	$10^{-2}$	50	15	44.3

<sup>1</sup> Sample ruptured; strain is approximately the point at which load dropped abruptly.

crostructure, the sections were “welded” to glass plates using a thin bead of water around the perimeter, rather than melted on. Photographs were taken under several different lighting conditions, as discussed in Melanson and others (1999). Grain-size was determined using the intercept method (Dieter, 1976) for a number of samples.

**RESULTS**

**Mechanical behavior**

Figure 1 shows an example of the axial strain, confining pressure and corrected deviatoric stress measured during a complete test sequence. The confining pressure was slowly increased and held, then the deviatoric stress was applied quickly and held until a specified limit of axial shortening was reached. Figures 2–8 show the axial strain with time (view a) and the strain rate plotted as a function of total strain (view b) for only the creep portion of all tests. Three factors are highlighted by these plots: the time prior to accelerated creep (view a), and the creep rate and stability or failure under load (view b).

Figures 2 and 3 demonstrate the effects of varying the deviatoric stress for a selected confining pressure ( $P_c = 20$ ; Fig. 2) and for a selected hydrostatic stress ( $p = 55$  MPa; Fig. 3). The effects of an increased deviatoric stress are to decrease the time prior to accelerated creep and to increase the tertiary creep rate, at both lower and higher confining pressures. In most of the samples shown in Figures 2–8, creep rates showed a rapid increase within the first 10% strain, and then increased much more slowly with continuing deformation.

The exceptions to this pattern were, in every case, samples which ruptured macroscopically during testing, as shown in curve 4 of Figure 3b. Rupture is defined, for our purposes, as failure of the specimen along a through-going fault plane. (See Figure 11 for a schematic representation, and Plate 10\* for examples of thin sections through ruptured specimens.) The generic term “rupture” is used because the faulted specimens often show evidence of both brittle fracture and ductile deformation; the contributions of each mechanism to the ultimate failure cannot be clearly distinguished in our samples. Curves for samples which ruptured are marked with an asterisk.

\* For colour plates see section preceding this paper.

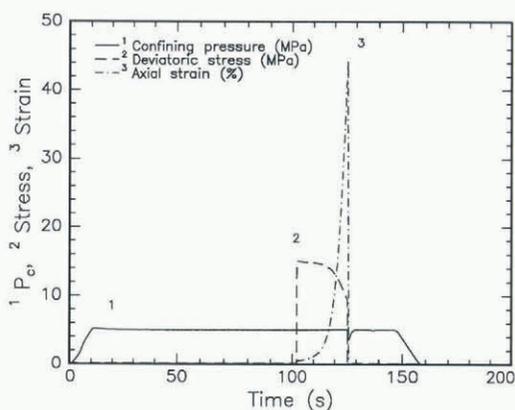


Fig. 1. Example of the corrected deviatoric stress, confining pressure and strain curves from a typical creep test.

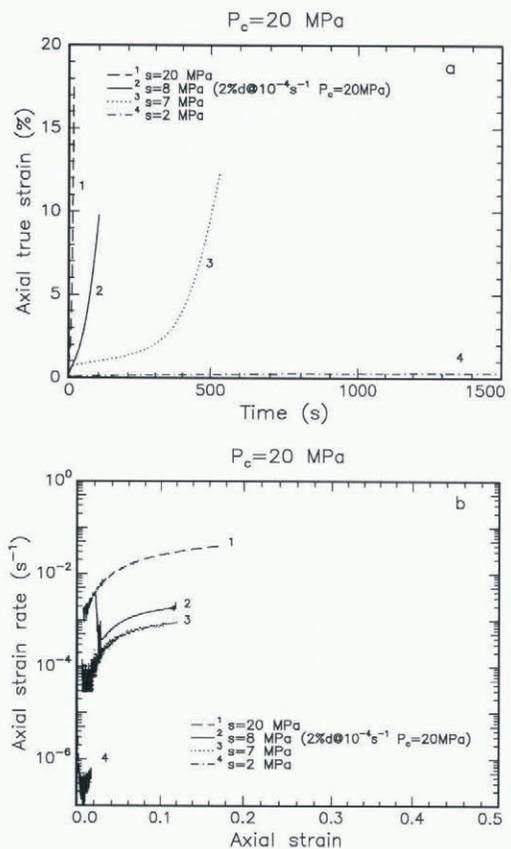


Fig. 2. Axial true strain as a function of time (a), and strain rate as a function of total cumulative axial strain (b), plotted for four tests done with a confinement  $P_c = 20$  MPa and deviatoric stresses as shown. Sample subjected to prior damage is noted.

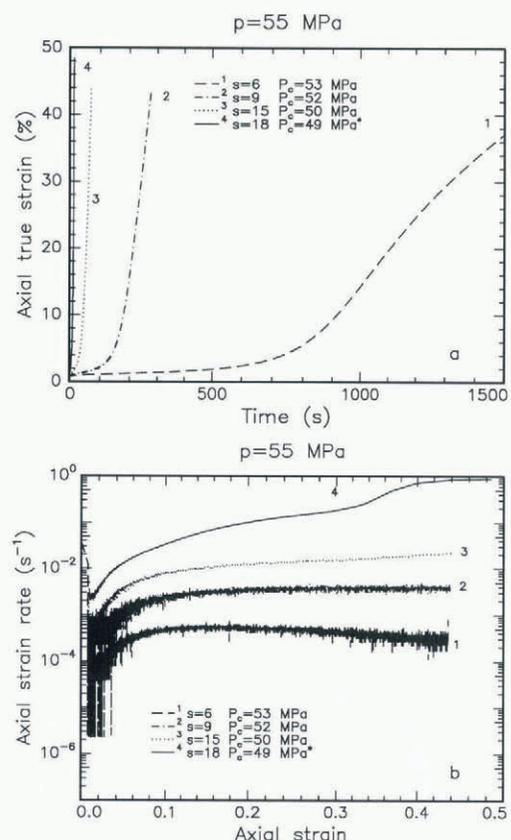


Fig. 3. Axial true strain (a) and strain rate (b) plotted for four tests done at a hydrostatic stress  $p = 55$  MPa and deviatoric stresses as shown. Curve marked with \* asterisk is from a sample which ruptured.

The tests done at a deviatoric stress of  $s = 15$  MPa and confining pressures  $P_c = 5$  and 50 MPa are plotted in Figures 4 and 5, respectively. All remaining tests at  $s = 15$  MPa and various confinements are plotted in Figure 6. Figure 7 shows curves for four tests done using the very highest deviatoric stresses. Figure 8 shows the creep tests of Melanson (1998) in which samples were subjected to a prior damage step of 12% axial strain.

In general, the tests at  $P_c = 5$  MPa show the onset of accelerating creep within 10–15 s following application of the deviatoric stress (Fig. 4a). Deformation was extremely rapid after that point, with creep rates on the order of  $5$  to  $9 \times 10^{-2} \text{ s}^{-1}$ . Curve 2 is the sample that was subjected to a pressure drop during loading; its behavior is indistinguishable from samples which had the axial load applied directly.

At confining pressures of  $> 5$  MPa, there was more variation in both the time to reach accelerated creep and the creep rate. Eight tests were done at  $P_c = 50$  MPa using the standard granular ice (Fig. 5). Some of these tests (curves 1–3) failed along a rupture plane; the strain rates do not stabilize at higher strains and the ice did not sustain the applied load (Fig. 5b). The samples which did not rupture (curves 4–8) reached a relatively constant strain rate ( $1$  to  $3 \times 10^{-2} \text{ s}^{-1}$ ) after a slightly longer period of time.

Three additional tests plotted in Figure 5 were done under the same stress conditions using a fine-grained sample (curve 9), a large-grained sample (curve 10) and a standard ice sample that was subjected to a prior damage step (curve 11). The fine-grained specimen behaved similarly to the standard grain-size specimens that did not rupture. The large-grained sample experienced only a small amount of strain before rupturing. The damaged sample had previously been subjected to 18% strain using a deviatoric stress of 15 MPa at a confinement of 5 MPa (shown as curve 5 in Figure 3). The impact of this damage on the subsequent creep behavior at

50 MPa confinement is dramatic. The sample had reached the stage of accelerated creep during the initial loading, with a peak strain rate on the order of  $3 \times 10^{-2} \text{ s}^{-1}$ . In the subsequent loading at higher confinement, the strain rate was immediately very high ( $2 \times 10^{-2} \text{ s}^{-1}$ ) without a stage of slow deformation, and the specimen ultimately ruptured.

The remaining specimens tested at  $s = 15$  MPa and various confinements (Fig. 6) were, in general, slower to reach accelerated creep than samples tested at  $P_c = 5$  MPa. Tests with  $P_c$  in the range 30–40 MPa (curves 4–6) tended to behave more like the samples tested at  $P_c = 50$  MPa which did not rupture, with a longer time to accelerated creep and a lower creep rate ( $1$  to  $3 \times 10^{-2} \text{ s}^{-1}$ ). Two tests in which the ram made relatively hard contact during application of the creep load (marked with a tilde) do not appear to be significantly different from the rest. The tests at the highest confining pressures shown ( $P_c = 45$  and 60 MPa; curves 1 and 3) both had a very rapid creep rate which continued to increase with axial strain; both these samples ruptured fairly late in the deformation. With the sample tested at  $P_c = 60$  MPa, for example, the two sections of ice were fused together when the sample was removed from the confining cell (see Plate 10); only about 1 cm of offset had occurred.

Four tests subjected to very high deviatoric stresses are plotted in Figure 7. One specimen was subjected to a damage step of 4% axial shortening at a strain rate of  $10^{-4} \text{ s}^{-1}$  with a confinement of 20 MPa prior to the application of the creep load. This sample and the two specimens tested at  $P_c = 20$  MPa and  $s = 20$  MPa reached accelerated creep very rapidly, and have virtually identical strain-rate vs strain curves (Fig. 7b). Two samples ruptured near the end of the test

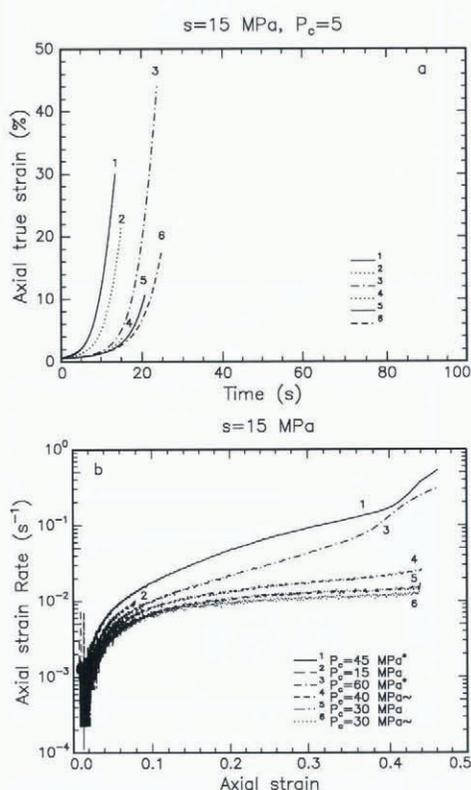


Fig. 4. Axial true strain (a) and strain rate (b) plotted for six tests done at a deviatoric stress  $s = 15$  MPa with confinement  $P_c = 5$  MPa.

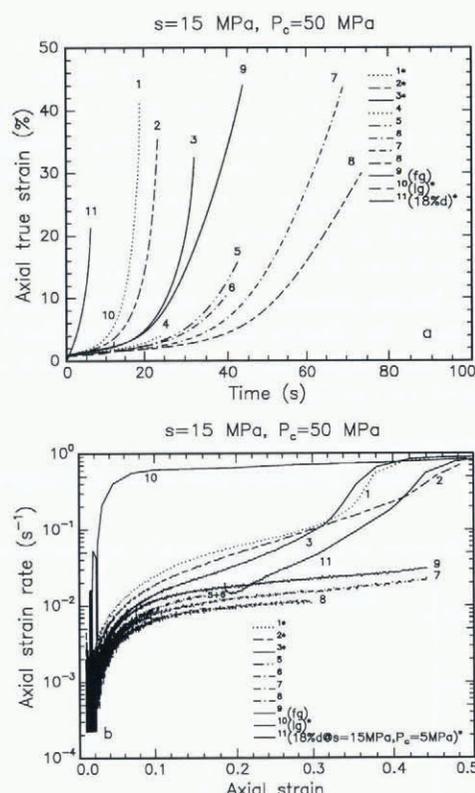


Fig. 5. Axial true strain (a) and strain rate (b) plotted for 11 tests done at a deviatoric stress  $s = 15$  MPa with confinement  $P_c = 50$  MPa. Curves marked with asterisk are from samples which ruptured. Curves marked “lg” and “fg” are from initially large- and fine-grained samples; for the curve marked “18% d” the pressure was increased from 5 MPa to 50 MPa after an initial 18% axial strain.

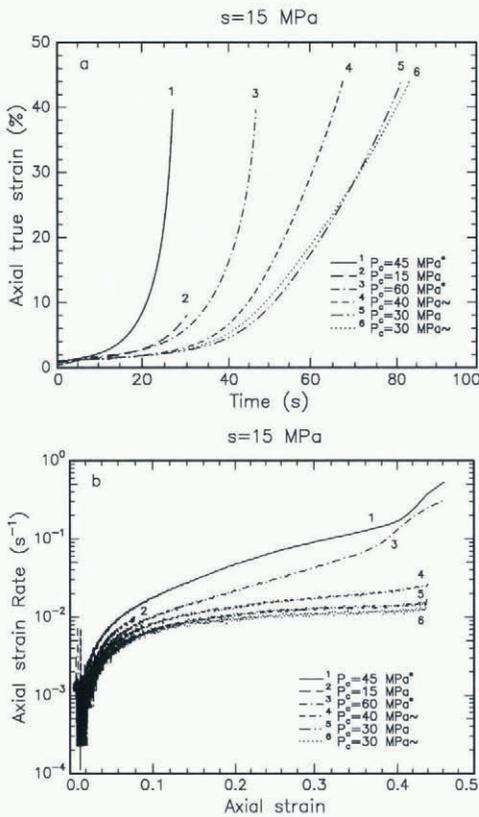


Fig. 6. Axial true strain (a) and strain rate (b) plotted for six tests done at deviatoric stress  $s = 15$  MPa with confinement between  $P_c = 15$  MPa and 60 MPa. Curves marked with an asterisk are from samples which ruptured. Curves marked with a tilde show hard contact with ram during initial application of creep load.

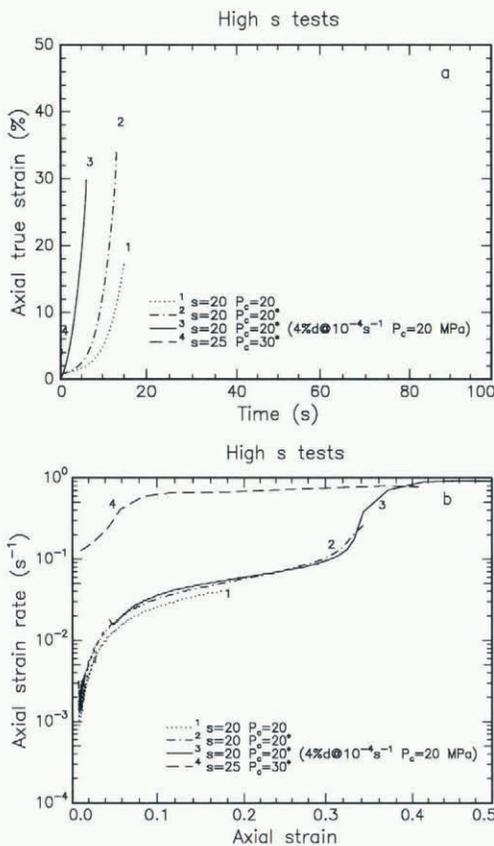


Fig. 7. Axial true strain (a) and strain rate (b) plotted for four tests done using high deviatoric stress  $s = 20$ – $25$  MPa. Curve marked “4% d” is for the sample subjected to a damage step prior to creep loading.

(curves 2 and 3). The specimen tested at  $P_c = 30$  MPa and  $s = 25$  MPa ruptured in  $< 2$  s (curve 4).

Six specimens were subjected to a damage step of 12% axial strain at constant strain rates prior to the application of the creep load (Fig. 8a). Like the damaged sample shown in Figure 5 (curve 11), these samples have extremely high strain rates immediately upon application of the creep load. However, the creep rates are comparable with the initially undamaged samples (i.e. samples subjected only to creep loading) when the rates are compared at the equivalent level of total axial strain.

This consistency in the creep rates is evident in Figure 9, which shows strain rates plotted against the corrected deviatoric stress at a number of axial strain levels for all of the test data presented, including the constant-strain-rate deformation steps on the damaged samples. Strain rates for samples that ruptured are shown only for creep prior to the loss of strength; beyond that point, the rate reflects the limit of the ram acceleration. Linear least-squares regressions were done for the data at strain levels 2%, 5%, 10%, 20% and 30%. Because most of the samples taken to the higher axial strain levels were subjected to a deviatoric stress of 15 MPa, the regression lines are shown for general comparison. The data taken at a given level of strain fit an equation of the form:

$$\dot{\epsilon} = \frac{s^n}{\nu},$$

where  $\nu$  is the effective viscosity and  $n$  characterizes the dependence of the strain rate ( $\dot{\epsilon}$ ) on the deviatoric stress ( $S$ ). The values of these parameters are given in Table 3. Although there are fewer data available at high strains, the overall trend of the values  $\nu$  and  $n$  is consistent with the raw data. The effective viscosity decreases by more than an order of magnitude with the initial 10% strain, during which the most

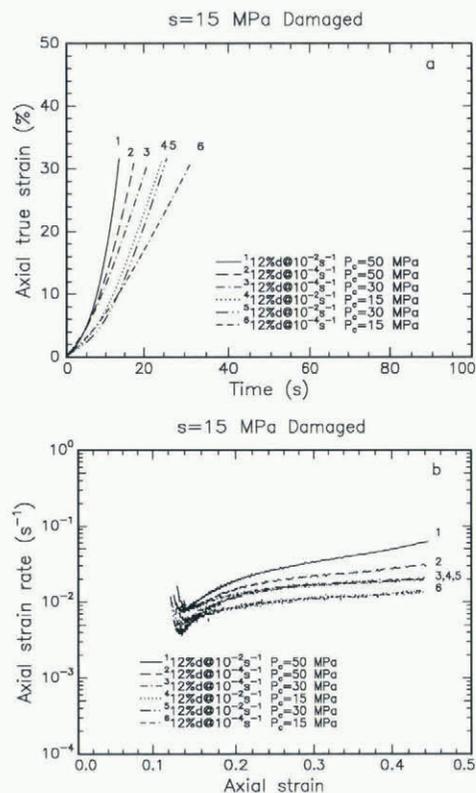


Fig. 8. Axial true strain (a) and strain rate (b) plotted for six tests which were subjected to a prior damage step of 12% axial strain at constant strain rates as given. These creep tests were done at a deviatoric stress  $s = 15$  MPa with  $P_c = 15$ , 30 or 50 MPa.

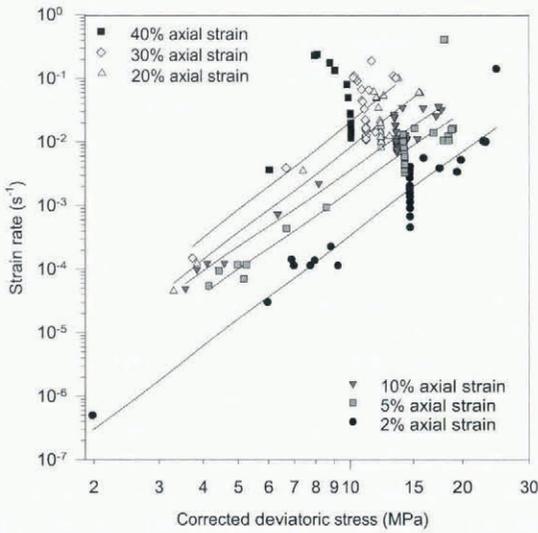


Fig. 9. Axial strain rate vs the corrected deviatoric stress plotted for all tests. Strain rate was determined at five levels of axial strain.

substantial microstructural changes occur in the ice. At higher strain levels, the effective viscosity stabilizes. Although creep rates increase by several orders of magnitude with increasing axial strain, the stress exponent *n* remains within the range 4.0–4.6 over the full range of strains.

The significance of the deviatoric stress in the creep behavior of ice is also seen in the time required for the sample to reach accelerated creep, as shown in Figure 10. For our purposes, a measure of the time to accelerated creep was derived from a line fit to the strain–time curves at 5–10% strain; the time-intercept of this line was used. This strain interval was chosen because in general all the samples show

Table 3. Effective viscosity ( $\nu$ ), stress exponent ( $n$ ) and correlation coefficient ( $r^2$ ) derived from the strain rate and corrected deviatoric-stress data at selected levels of cumulative axial strain

Total axial strain	$\nu$	$n$	$r^2$
% True strain	MPa <sup>-n</sup> s		
2	10 <sup>7.3</sup>	4.4	0.8934
5	10 <sup>6.9</sup>	4.1	0.8136
10	10 <sup>6.4</sup>	4.0	0.9485
20	10 <sup>6.5</sup>	4.4	0.8385
30	10 <sup>6.3</sup>	4.6	0.5813

a dramatic increase in strain rate above approximately 5% strain. Although the choice is somewhat arbitrary, it allows for objective determination of a time (rather than identification of the inflection by eye), it allows us to compare most of the test data (since some samples were taken only to intermediate strain levels) and it yields results consistent with qualitative observations of the raw data. These results show a strong correlation between deviatoric stress and the time to accelerated creep. The regression line is fit only to the data from initially intact ice samples:

$$t_{ac} = \frac{10^{5.88}}{S^{3.96}} \text{ with correlation coefficient } r^2 = 0.8540$$

where  $t_{ac}$  is our measure of the time to accelerated creep. The

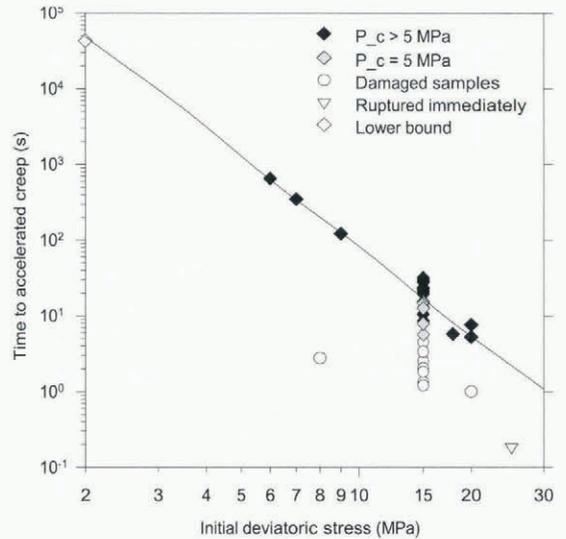


Fig. 10. Time to accelerated creep vs the initial (i.e. uncorrected) deviatoric stress for all tests. Linear regression was done only with data marked by filled symbols; these are distinguished at 15 MPa for low and high confining-pressure tests. Open circles are samples subjected to a prior damage step. Open diamond reflects a lower bound, as discussed in the text.

data point at 2 MPa was excluded from the regression because this sample reached only 2% strain in approximately 43 000 s. It represents a lower bound on the measure we are using.

### Microstructure

Schematic representations of the macroscopic shapes of typical deformed samples are shown in Figure 11. Samples that deformed without rupture showed axial shortening and lateral expansion which was relatively uniform along the length of the sample (Fig. 11a); often ice had extruded around the end platens. Samples that ruptured after a significant period of creep generally showed both lateral strain (similar to the intact creep samples) and a distinct failure plane at approximately 45° to the axial loading direction (Fig. 11b). These samples did not show uniform lateral expansion; often they had a prominent bulge near the center, indicative of strain localization. Samples that failed rapidly following the application of the creep load had a rupture plane oriented at approximately 45° to the axial load direction (Fig. 11c). These samples showed no appreciable lateral expansion, consistent with the fact that they ruptured at very low axial strain levels.

Photographs of thin sections from some of the test speci-

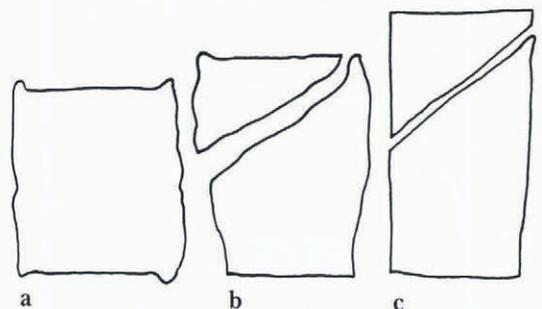


Fig. 11. Schematic representation of macroscopic shape of deformed samples following testing. Most samples deformed relatively uniformly (a). Samples which ruptured typically showed a combination of lateral strain and faulting (b). In two cases, samples failed rapidly without significant lateral strain (c).

mens are shown in Plates 4–10. Refer to Plate 1 for a view of a horizontal thin section through an undeformed specimen. In general, the grain boundaries are straight and regular prior to deformation, and few bubbles or flaws are visible.

Plates 4–10 show photographs of both horizontal and vertical thin sections of selected specimens following testing. Sections are shown which contrast the progressive microstructural change with strain for tests with a deviatoric stress of 15 MPa at two confining pressures,  $P_c = 5$  MPa (Plates 4–6) and  $P_c = 50$  MPa (Plates 7–9).

At the lowest confining pressure, the first increment of strain (4%) causes the formation of both intra-granular and inter-granular microcracks (Plates 4–6, view a). Most of the original grains are preserved, with only localized formation of fine-grained material in areas where extensive cracking has occurred. Intra-granular cracks tend to be highly aligned within a given grain (Plate 4), possibly reflecting the strike of the basal plane. Some grains have more than one set of oriented cracks within them. With increasing strain (Plates 4–6, view b), crack density within grains increases, and these cracks tend to remain strongly aligned.

At the maximum strain (44%) (Plates 4–6, view c), few original grains remain intact. Between crossed polarizing filters, the sample appears as a mass of extremely fine-grained material surrounding a few large grains. Many of these large grains have a preferred alignment of  $c$  axes parallel to the loading direction (inferred from persistent optical extinction in horizontal thin sections as they are rotated). The addition of side lighting reveals that the fine-grained material corresponds with local dense concentrations of highly aligned cracks. Although this specimen did not rupture, the vertical section (Plate 6c) shows evidence of some localization on a plane, visible in the bulge at the lower right and upper left corners (relative to the sample axis). There is remarkable uniformity of texture throughout the specimen, even in the ice that has extruded around the platen. The grains directly adjacent to the platen remain slightly larger (possibly a result of non-uniform stress distribution), but this zone is only a few grain diameters wide.

Increasing the confining pressure to 15 MPa or higher results in suppression of microcracks as a dominant microstructural element. The sections shown in Plates 7–9 are typical of the non-ruptured samples from tests with  $P_c > 5$  MPa. Some cracking is visible in the sample subjected to 4% strain (Plates 7–9, view a), but the crack density is significantly lower than in the specimens tested at low confinement. Fine-grained material forms initially at the grain boundaries. With increasing axial strain the fraction of fine-grained material increases (Plates 7–9, views b and c). The size of the fine grains is slightly larger than in the highly microcracked samples (Plates 4–6), and the grain boundaries are straight and regular. However, the texture consisting of isolated large grains in a fine-grained matrix is similar. In some cases, fine-grained material is clearly associated with cracks (as in the large black grain in Plate 8b). In general, however, these grains are uncracked.

Macroscopic rupture occurred in a number of the specimens subjected to very high deviatoric stresses ( $s > 15$  MPa) at confinements of  $> 15$  MPa. Two such samples are shown in Plate 10. The sample tested at  $P_c = 60$  MPa and  $s = 15$  MPa is shown in a vertical thin section (Plate 10a). Rupture occurred along a plane at approximately  $45^\circ$  to the axis, visible as the protrusion on the left edge and the bright cracks in the center. The two portions of the sample were fused to-

gether when the triaxial cell was opened. Slip along this plane is believed to have occurred only during the last stages of strain, as indicated by the small amount of offset visible along the fault and by the strain-rate history (Fig. 6b). The grain texture of this sample is similar to others deformed under high confinement, with a mixture of large and fine grains and very few cracks. When viewed with side lighting, only a few small cracks are visible along the trace of the fault plane itself. In contrast, a sample which ruptured under conditions  $P_c = 30$  MPa and  $s = 30$  MPa after only a small amount of strain (a few per cent) has extensive microcrack damage to grains throughout, and a zone of fine-grained material is adjacent to the fault plane (Plate 10b and c).

Average grain-sizes are plotted in Figure 12 as a function of the total axial strain to which the sample was subjected during testing. (Note: not all the samples were analyzed for grain-size.) The average grain diameter remains relatively high for the first few per cent of axial strain. It decreases significantly by 10% axial strain and then tends to stabilize with increasing deformation. Low ( $P_c = 5$  MPa) and high ( $P_c > 5$  MPa) confinement tests are differentiated, as are the very highest deviatoric stress tests. The scatter in values from the undeformed samples indicates the variability of the starting material, rather than error in the measurements. The fine-grained samples marked by solid squares were made using snow as the seed ice. Though initial grain-size of these samples was smaller and more uniform than that of the standard specimens, the grain-size of the deformed sample does not differ significantly.

## DISCUSSION

We consider the mechanical behavior of the ice in relation to the microstructural changes observed in thin section, with a caution that the sections may not entirely reflect the state of the ice while it is under load. (The release of stress at the end of the test, and the time lag prior to thin-sectioning, may cause some additional microstructural changes, such as the quenching of melted regions, stress-relief cracking, static recrystallization, grain growth or crack healing.)

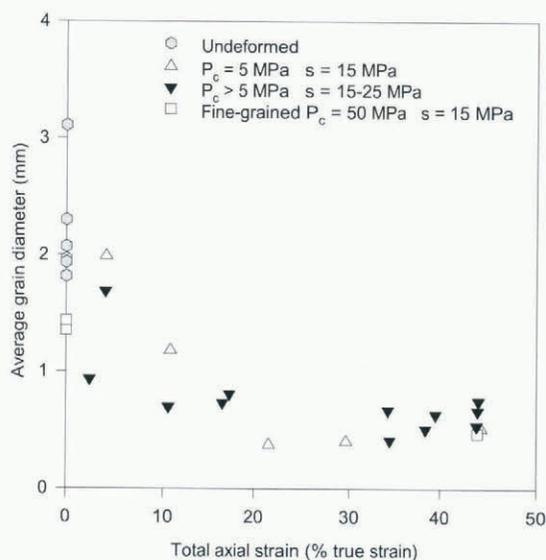


Fig. 12. Plot of average grain diameter as a function of the total cumulative axial strain (in % true strain).

## Microstructure

Much of the microstructural change in the initial stage of deformation is associated with grain boundaries, by cracking at low confinement and by nucleation of new grains at higher confinement. Grain-size decreases most dramatically during the first 10% axial strain. At high strains, the texture (consisting of a fine-grained mass surrounding isolated, large grains) is common to samples tested at all levels of confinement. Only the two specimens that ruptured after a very small amount of axial strain retained a strong signature of the original grain texture. We interpret the large grains as remnants which survived the initial breakdown of ice structure, possibly because they were not favorably oriented for either dense fracturing or recrystallization. As deformation proceeded and the surrounding material softened, the resolved shear stress on these large grains could in fact be quite small, allowing their preservation even with considerable sample strain.

Although the apparent mechanism by which the fine-grained material formed is quite different at low and high pressures (dense cracking vs dynamic recrystallization), there is no strong confining pressure dependence observed in the average grain-size (Fig. 12). Rather, the grain-size tends to be consistently related to the total accumulated axial strain. This conclusion is supported by work of Melanson and others (1999) in which samples were subjected to constant-strain-rate tests rather than constant-load tests. Those tests show a dependence of average grain-size on total axial strain similar to that in the creep tests presented here. Results from both these studies demonstrate that during the first few per cent of axial strain the most dramatic microstructural changes occur, the ice passes through its peak strength, the strain rates are at their lowest and the greatest variability in strain rates is observed.

## Mechanical behavior

At all confining pressures, creep rates during the first few per cent of strain were relatively low. The differences in these rates among samples tested under the same nominal conditions may be a function of microstructural differences in the undeformed specimens, differences in test conditions such as confinement or temperature, or the occurrence of localization along rupture planes in some samples. However, the variations in time to accelerated creep among, for example, the samples tested at 15 MPa deviatoric stress are relatively small compared with the substantial influence of the deviatoric stress itself. The dramatic effects of prior damage are also evident, in the very short times within which the damaged samples reach accelerated creep.

In this work, results have been compared for samples that were subjected to several different loading paths: deformation by creep; deformation by creep following a damage step at a constant rate; and deformation at a constant axial strain rate (Melanson and others, 1999). With only a few exceptions, creep rates were remarkably similar among samples at specific levels of strain for a given deviatoric stress. They were relatively insensitive to how that level of strain had been attained. This leads us to conclude that the total strain accumulated during the prior load history of a sample is a strong factor in determining the creep rate at a given deviatoric stress. This conclusion is consistent with results from uniaxial tests on granular fresh-water ice by Cole (1983) at lower strain rates and stresses, and with the

work of Jacka (1984), Jacka and Maccagnan (1984) and Jacka and Li (1994), which demonstrated that an equilibrium grain-size and a grain fabric develop during tertiary creep of ice deformed to high strains in uniaxial tests. The final grain-size can represent a reduction or an increase in average grain-size, depending on the initial ice structure and test conditions. A characterization of previous stress history that focused on the effect of hydrostatic pressure during the test was given in Jordaan and others (1999). This indicated an initial hardening with pressure related to suppression of cracking, and a subsequent softening that is believed to be related to the effect of pressure melting. Though we have not focused here on the pressure-dependence of creep rates, their findings are consistent with the present test results.

In the present study, the strain rate in a given sample increases by two to three orders of magnitude as the total axial strain increases. This is reflected in the mechanical properties of the test specimens by the drop in effective viscosity within the first 10% strain, correlating with the dramatic microstructural changes visible in thin sections. Yet, there is remarkably consistent power-law behavior among the entire suite of samples. The material does continue to soften (seen in the viscosities, which continue to decrease, and the creep rates, which continue to increase with strain), but the most dramatic softening corresponds to the initial breakdown of grain structure. In effect, the ice undergoes a transformation to a material with a new structure and different properties that responds to the stress conditions in a relatively predictable way.

The most significant exceptions to this pattern were the samples that ruptured. Among samples tested under the same stress conditions, the difference was, in most cases, evident early in the strain history. The samples which eventually ruptured tended to have higher strain rates early on, suggesting that localization associated with rupture began during the initial breakdown of the ice structure. Relatively small variations in the test temperature, differences in the magnitude of stress concentrations, or variability in the flaw or grain structure of the undeformed specimens may act alone or in combination to weaken the material at this stage of the deformation.

## Deformation mechanisms/damage

Several deformation mechanisms are known to influence the mechanical behavior of ice under a range of load conditions. The importance of cracking, dynamic recrystallization and pressure melting in the deformation of ice has long been recognized (Glen, 1955; Barnes and Tabor 1966; Jonas and Müller 1969; Kheisin and Cherepanov, 1973; Jones, 1982; Jones and Chew 1983; Glen and Ives, 1988; Xiao and others 1993).

In broad terms one may conclude on the basis of our results that cracking is the predominant damage mechanism at the lowest confining pressure, and dynamic recrystallization and melting at the higher pressures. However, observations of microstructural changes demonstrate that all three mechanisms combine under a wide range of load conditions to accommodate deformation. At low confinements, where cracks are a dominant microstructural element, grain-size reduction is not exclusively defined by the formation of cracks. The formation of new, smaller grains is intimately associated with both isolated cracks and dense arrays of cracks, but fine-grained regions that are clear of cracks are also found. The consistency in the constitutive behavior suggests that dynamic recrystallization may be acting in conjunction with the crack-

ing even at low confinements to enhance creep rates. Furthermore, melting in ice under load can result from both pressure melting and frictional heating, as observed by Gagnon (1994) during laboratory compression tests. Melting may therefore also play a role in deformation at low confinements, where pressure melting would not be expected.

At the higher confining pressures, persistence of a few isolated cracks to very high strains within the large grains might suggest that some evidence would be seen if extensive cracking were occurring. Cracking followed by rapid healing may be occurring continuously within the fine-grained mass while these specimens are under load, particularly if local melt is also present. The release of stress and the quenching of specimens at the end of tests would obscure evidence of this process. Clearly the texture of these specimens is markedly different from that of specimens deformed at low confinement. Only direct observation of the samples under load (as done at lower confinements by Rist and others, 1994, for example) would establish conclusively that cracking is *not* occurring at the higher confinements.

Indirect evidence suggests that at the highest confinements, melting contributes to the deformation. First, the samples tested at high confinements were in some cases perfectly welded across the rupture plane by the time the sample was retrieved from the confining cell. Second, reducing the temperature by only 1° or 2°C for some of the higher-confinement tests resulted in deformation without rupture and lower creep rates. Third, the sample tested first at 5 MPa and then at 50 MPa confinement contained regions that were extremely fine-grained but perfectly clear, despite the fact that the initial deformation at low confinement would have caused extensive cracking. The occurrence of local melting under higher-confinement conditions is one explanation for all these observations.

We conclude therefore that the “damage” which results in the softening of these specimens cannot be quantified simply as, for example, a density of cracks. Rather, the softening may be more appropriately related to the deformation process that is activated while the specimen is under load. Elements such as the volume of grain boundary, the dislocation density and distribution, the density of transient cracks which occur and then heal under load and the volume of melt at the grain boundaries may all constitute damage. However, they are not easily quantified in thin sections at ambient pressure and therefore cannot be related in a straightforward manner to the dynamic condition of the sample while it is under load.

The behavior of ice has strong similarities with the deformation of metals at high homologous temperatures, due to the prevalence of dynamic recrystallization as a deformation mechanism (Bailey and Hirsch, 1961; Sellars, 1978). The results from our tests are also consistent with observations of strain localizations and brittle/ductile behavior at both large and small scales in rocks (Derby, 1990). Mylonites, for example, are fine-grained rocks which occur localized in fault zones, are soft or weak relative to the surrounding intact rock and are believed to accommodate large shearing strains in fault zones (Etheridge and Wilkie, 1979). There are many analogous deformational features seen in these laboratory ice samples. In field examples of deformation localization, evidence is present that the material within shear zones has experienced sufficient frictional heating to cause melting and extrusion of molten material (Hobbs and others, 1976; Scholz, 1990). Furthermore, a de-

formation mechanism analogous to pressure melting in ice is pressure solution in other rocks, in which local high stress concentrations enhance the chemical dissolution of material in the presence of pore fluids (Raj, 1982). Therefore, studies of microstructural change in ice may have much wider implications for understanding the mechanisms of deformation and metamorphism in a variety of materials.

### Significance in ice–structure interaction

Thin sections of iceberg ice deformed in field-scale impact test (Muggeridge and Jordaan, 1999) show extensive damage in the form of ice fragments, often rotated and offset along localized rupture zones, and larger zones that contain primarily fine-grained material. The microstructural changes observed in these samples, deformed under true impact conditions, are strikingly similar to those produced in some of the laboratory tests presented here.

The ejection of fine-grained crushed material from the contact zone has been observed during some field impact and indentation tests. While this crushed material is softer than intact ice, the work of Singh and others (1995) and the results presented here indicate that even soft, highly damaged ice is capable of sustaining, and therefore transmitting, high loads for short periods of time. In the context of ice–structure interaction, a significant amount of damage can be done to a vessel or structure in a short period of time by impact with an iceberg. Because of the complexity of this problem, the approach to modeling ice behavior during ice–structure interaction has involved deriving a workable constitutive relation from these and other laboratory tests, and then implementing it in finite element simulations of contact zones (Jordaan and others, 1999).

### CONCLUSIONS

This experimental study was undertaken to characterize the mechanical behavior and microstructural changes in ice subjected to stress conditions present in an ice–structure interaction zone. Results of triaxial creep tests under confining pressures up to 60 MPa, deviatoric stresses up to 25 MPa and axial strains up to 44% have been presented. Microstructural characterization focused on those tests using a deviatoric stress of 15 MPa at low (5 MPa) and high (50 MPa) confinement.

In general, samples tested at a confining pressure of 5 MPa failed rapidly without rupture and contain dense microcrack populations. Samples tested at higher confining pressure deformed more slowly if the sample remained intact, or very quickly if the sample failed along a macroscopic rupture plane. In all cases, if the sample experienced more than a few per cent strain, substantial microstructural change was visible in thin sections. The original texture was replaced by a matrix of fine-grained material surrounding larger grains interpreted as remnants of the original grains. In samples tested at 5 MPa confinement, this fine-grained matrix was highly microcracked; in samples tested at high confinement, cracks were scarce.

The constitutive behavior of this ice obeys a power law when the strain rates are correlated with the deviatoric stress at the same levels of cumulative axial strain. The effective viscosity decreases from  $10^{7.8}$  to  $10^{6.4}$  with the initial 10% strain, correlating with the most substantial microstructural changes in the ice. At higher strain levels, the effective viscos-

ity reaches a relatively stable value. The exponent representing the deviatoric stress dependence remains within the range 4.0–4.6. This behavior is notable given that the apparent deformation mechanism is quite different at low and high confinements. The time to accelerated creep, being another measure of the creep rate at low strains, also correlates very strongly with the deviatoric stress. Factors such as the temperature and confining pressure contribute to the variation in sample behavior at a given deviatoric stress. The exceptional cases were those samples which ruptured immediately, and those which had been subjected to a damage step prior to application of the creep load. In both cases these samples reached accelerated creep very quickly.

Several potential deformation mechanisms exist in ice under these conditions, including microcracking, pressure melting, melting caused by frictional heating, and recrystallization-enhanced creep. Microstructural observations suggest that all these elements are active over a range of stress conditions, though the relative importance of each is a function of the confinement, deviatoric stress level, total accumulated strain and temperature.

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## REFERENCES

- Bailey, J. E. and P. B. Hirsch. 1961. The recrystallization process in some polycrystalline metals. *Proc. R. Soc. London, Ser. A*, **267**, 11–30.
- Barnes, P. and D. Tabor. 1966. Plastic flow and pressure melting in the deformation of ice I. *Nature*, **210** (5039), 878–882.
- Cole, D. M. 1983. The relationship between creep and strength behavior of ice at failure. *Cold Reg. Sci. Technol.*, **8** (2), 189–197.
- Derby, B. 1990. Dynamic recrystallization and grain size. In Barber, D. J. and P. G. Meredith, eds. *Deformation processes in minerals, ceramics and rocks*. London, Unwin Hyman, 354–364.
- Dieter, G. E. 1976. *Mechanical metallurgy*. New York, McGraw-Hill.
- Etheridge, M. A. and J. C. Wilkie. 1979. Grain size reduction, grain boundary sliding and the flow strength of mylonites. *Tectonophysics*, **58**, 159–178.
- Frederking, R., I. J. Jordaan and J. S. McCallum. 1990. Field tests of ice indentation at medium scale, Hobson's Choice ice island, 1989. In *IAHR Symposium on Ice, Espoo, Finland, August 20–24, 1990. Proceedings. Vol. 2*. Espoo, Helsinki University of Technology, 931–944.
- Gagnon, R. E. 1994. Generation of melt during crushing experiments on freshwater ice. *Cold Reg. Sci. Technol.*, **22** (4), 385–398.
- Gagnon, R. E. and P. H. Gammon. 1995. Triaxial experiments on iceberg and glacier ice. *J. Glaciol.*, **41** (139), 528–540.
- Glen, I. F. and H. Blount. 1984. Measurement of ice impact pressures and loads onboard CCGS *Louis St. Laurent*. In Lunardini, V. J., ed. *Third International Offshore Mechanics and Arctic Engineering Symposium (OMAE)*, New Orleans, Louisiana, February 12–16 1984. *Proceedings. Vol. 3*. New York, American Society of Mechanical Engineers, 246–252.
- Glen, J. W. 1955. The creep of polycrystalline ice. *Proc. R. Soc. London, Ser. A*, **228** (1175), 519–538.
- Glen, J. W. and D. J. Ives. 1988. Fish antifreeze proteins and the creep of polycrystalline ice. *J. Glaciol.*, **34** (118), 291–292.
- Hobbs, B., W. Means and P. Williams. 1976. *An outline of structural geology*. New York, Wiley.
- Jacka, T. H. 1984. Laboratory studies on relationships between ice crystal size and flow rate. *Cold Reg. Sci. Technol.*, **10** (1), 31–42.
- Jacka, T. H. and Li Jun. 1994. The steady-state crystal size of deforming ice. *Ann. Glaciol.*, **20**, 13–18.
- Jacka, T. H. and M. Maccagnan. 1984. Ice crystallographic and strain rate changes with strain in compression and extension. *Cold Reg. Sci. Technol.*, **8** (3), 269–286.
- Jonas, J. J. and F. Müller. 1969. Deformation of ice under high internal shear stresses. *Can. J. Earth Sci.*, **6** (4, Part 2), 963–968.
- Jones, S. J. 1982. The confined compressive strength of polycrystalline ice. *J. Glaciol.*, **28** (98), 171–177.
- Jones, S. J. and H. A. M. Chew. 1983. Creep of ice as a function of hydrostatic pressure. *J. Phys. Chem.*, **87** (21), 4064–4066.
- Jones, S. J. and G. P. Johari. 1977. Effect of hydrostatic pressure on bubbles in ice. *International Association of Hydrological Sciences Publication 118* (Symposium at Grenoble 1975—Isotopes and Impurities in Snow and Ice), 23–28.
- Jordaan, I. J., D. G. Matskevitch and I. L. Meglis. In press. Disintegration of ice under fast compressive loading. *Int. J. Fract.*, **97** (1–4), 279–300.
- Kalifa, P., G. Ouillon and P. Duval. 1992. Microcracking and the failure of polycrystalline ice under triaxial compression. *J. Glaciol.*, **38** (128), 65–76.
- Kheisin, D. E. and N. V. Cherepanov. 1973. Change of ice structure in the zone of impact of a solid body against the ice cover surface. *Proble. Arct. Antarct.*, **33–35**, 239–245.
- Melanson, P. M. 1998. Damage and microstructural change in laboratory grown ice under high pressure zone conditions. (M. Eng. thesis, Memorial University of Newfoundland.)
- Melanson, P. M., I. L. Meglis, I. J. Jordaan and B. M. Stone. 1999. Microstructural change in ice: I. Constant-deformation-rate tests under triaxial stress conditions. *J. Glaciol.*, **45** (151), 417–422.
- Mellor, M. and D. M. Cole. 1983. Stress/strain/time relations for ice under uniaxial compression. *Cold Reg. Sci. Technol.*, **6** (3), 207–230.
- Muggeridge, K. J. and I. J. Jordaan. 1999. Microstructural change in ice: III. Observations from an iceberg impact zone. *J. Glaciol.*, **45** (151), 449–455.
- Raj, R. 1982. Creep in polycrystalline aggregates by matter transport through a liquid phase. *J. Geophys. Res.*, **87** (B6), 4731–4739.
- Rist, M. A. and S. A. F. Murrell. 1994. Ice triaxial deformation and fracture. *J. Glaciol.*, **40** (135), 305–318.
- Rist, M. A., S. A. F. Murrell and P. R. Sammonds. 1988. Experimental results on the failure of polycrystalline ice under triaxial stress conditions. In Saeki, H. and K.-i. Hirayama, eds. *IAHR88, 9th IAHR International Symposium on Ice, 23–27 August 1988, Sapporo, Japan. Proceedings. Vol. 1*. Sapporo, International Association for Hydraulic Research. Committee on Ice Problems, 118–127.
- Rist, M. A., S. J. Jones and T. D. Slade. 1994. Microcracking and shear fracture in ice. *Ann. Glaciol.*, **19**, 131–137.
- Scholz, C. 1990. *The mechanics of earthquakes and faulting*. Cambridge, Cambridge University Press.
- Sellars, C. M. 1978. Recrystallization of metals during hot deformation. *Philos. Trans. R. Soc. London, Ser. A*, **288**, 147–158.
- Singh, S. K., I. J. Jordaan, J. Xiao and P. A. Spencer. 1995. Flow properties of crushed ice. *J. Offshore Mech. Arct. Eng.*, **117** (4), 276–282.
- Sinha, N. K. 1979. Grain boundary sliding in polycrystalline materials. *Philos. Mag. A*, **40** (6), 825–842.
- Smith, T. R. and E. M. Schulson. 1993. The brittle compressive failure of fresh-water columnar ice under biaxial loading. *Acta Metall. Materialia*, **41** (1), 153–163.
- Stone, B. M., I. J. Jordaan, S. J. Jones and R. F. McKenna. 1989. Damage of isotropic polycrystalline ice under moderate confining pressures. In Axelson, K. B. E. and L. Å. Fransson, eds. *POAC 89. The 10th International Conference on Port and Ocean Engineering under Arctic Conditions, June 12–16 1989, Luleå, Sweden. Proceedings. Vol. 1*. Luleå, Tekniska Högskolan i Luleå, 408–419.
- Stone, B. M., I. J. Jordaan, J. Xiao and S. J. Jones. 1997. Experiments on the damage process in ice under compressive states of stress. *J. Glaciol.*, **43** (143), 11–25.
- Xiao, J., I. J. Jordaan and S. K. Singh. 1993. Pressure melting and friction in ice–structure interaction. In *IAHR 92. 11th International Symposium on Ice, Banff, Alberta, Canada, June 15–19, 1992. Proceedings. Vol. 3*. Edmonton, Alta, University of Alberta. Department of Civil Engineering, 1255–1268.

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